## Low-Temperature Chemical Vapor **Deposition of Ruthenium Dioxide from Ruthenium Tetroxide:** A Simple Approach to **High-Purity RuO<sub>2</sub> Films**

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## Introduction

Films of ruthenium dioxide are important for several electronic<sup>1</sup> and electrochemical<sup>2</sup> applications, and are usually obtained by solution methods from ruthenium-(III) chloride or other ruthenium compounds. Such films of RuO<sub>2</sub> are often contaminated with chloride impurities and show poor adhesion.<sup>1,2</sup> Recently, the preparation of high-quality RuO<sub>2</sub> thin films on silicon wafers has become of interest for electrode structures for developing new ferroelectric memories based on lead zirconate titanate (PZT).<sup>3</sup> Deposition of PZT films by a sol-gel process has been well established,<sup>4</sup> but since uniform, adherent RuO<sub>2</sub> films are difficult to prepare using such solution methods, the synthesis of RuO<sub>2</sub> films by chemical vapor deposition (CVD) is attractive. Pure ruthenium films have been prepared by CVD from several precursors, such as [Ru- $(acac)_3$ ], acac = acetylacetonate or its fluorinated derivatives, 5-7 [Ru<sub>3</sub>(CO)<sub>12</sub>], 5 [Ru(C<sub>5</sub>H<sub>5</sub>)<sub>2</sub>], 5.8 and [Ru(CO)<sub>4</sub>-(CF<sub>3</sub>C=CCF<sub>3</sub>)].<sup>9</sup> CVD of RuO<sub>2</sub> has been achieved previously from  $[Ru(C_5H_5)_2]$  or  $[Ru(acac)_3]$  in the presence of oxygen, but the films contain carbon impurities.5-8 Nevertheless, the  $RuO_2$  films prepared from  $[Ru(C_5H_5)_2]$ or [Ru(acac)<sub>3</sub>] in the presence of O<sub>2</sub> at 575 °C showed resistivities  $\rho = (8.99-64.3) \times 10^{-5} \Omega$  cm, which may be compared to that of RuO<sub>2</sub> single crystals [ $\rho = (3.5-5) \times$  $10^{-5} \Omega$  cm] and superior to those prepared from pyrolysis of ruthenium salts.<sup>2a</sup> CVD in the presence of hydrogen can often reduce carbon contamination, but this method cannot be used in this case since CVD from metallorganic ruthenium precursors in the presence of hydrogen gives ruthenium metal and not RuO<sub>2.5-9</sub> Clearly, for CVD of pure RuO<sub>2</sub>, a precursor which does not contain either carbon or halogen is desirable. This article reports that  $RuO_4$  is an excellent precursor for CVD of  $RuO_2$  and, since

solvent for	pressure carrier	substrate temp,	film morphology			XPS analysis <sup>a</sup>		
RuO4	gas	°C	substrate		adhesion	Ru	0	Cl
water	1 atm	150	glass	rough black	fair	18.7	81.3	
water	10 <sup>-1</sup> Torr	180	glass	rough black	poor	26.5	72.3	1.2*
water	1 atm	200	glass	rough black	poor	21.3	72.9	5.8
water	1 atm	150	Si(100)	rough black	fair	21.1	78.9	
CCL	1 atm/air	220	glass	shiny blue- black	good	25.8	64.7	9.5
CCL	1 atm/air	220	Si(100)	shiny blue- black	good	21.3	72.9	7.8
pentane	1 atm/air	150	glass	shiny black	poor	30.2	<b>69.</b> 8	

<sup>a</sup> Calculated values for RuO<sub>2</sub> are Ru, 33.3; O, 66.7. Values are given in atom percent. <sup>b</sup> Chlorine arises in nonreproducible amounts from impurities in commercial aqueous RuO<sub>4</sub>. No carbon was detected in any of the samples.

Table II. Atmospheric Pressure CVD of RuO<sub>2</sub> from RuO<sub>4</sub>

	substrate	film		XPS analysis <sup>a</sup>	
substrate	temp, °C	appearance	adhesion	Ru	0
glass	150	shiny dark-blue	excellent	26.4	73.6
Ši(100)	150	shiny purple	good	24.0	76.0
Al	150	shiny purple	excellent	25.0	75.0

<sup>a</sup> Analysis in atom percent. Theoretical values are Ru, 33.3; O, 66.7%.

the only other product is  $O_2$  (eq 1), the films are of high

$$RuO_4 \rightarrow RuO_2 + O_2 \tag{1}$$

purity. It has previously been noted that thermal annealing of RuO<sub>2</sub> films can lead to loss of ruthenium by disproportionation to  $Ru + RuO_4$ , which is volatile, and that vapor transport and crystal growth of RuO<sub>2</sub> at 1100 °C may involve the reversible formation of volatile RuO<sub>4</sub>.<sup>5,10</sup> The boiling point of RuO<sub>4</sub> is 129 °C (it melts at 27 °C), so it is much more volatile than metallorganic precursors.<sup>10-14</sup>

RuO<sub>4</sub> [caution]<sup>15</sup> is obtained by oxidation of ruthenium salts under basic conditions<sup>10</sup> and is commercially available in aqueous solution (Strem Chemicals). Solutions of  $RuO_4$ in CCl<sub>4</sub> or pentane, which have limited thermal stability, were easily obtained by solvent extraction and pure RuO<sub>4</sub> was obtained from the pentane solution by removing the solvent under vacuum (10<sup>-3</sup> Torr, ca. -60 °C). Using solutions of RuO<sub>4</sub>, CVD was carried out in a vertical coldwall reactor at atmospheric pressure with a substrate temperature of 150-220 °C, using a carrier gas of air or nitrogen for the solutions in CCl<sub>4</sub> or pentane (Table I), while CVD from pure RuO<sub>4</sub> was performed using a horizontal hot-wall reactor at 150 °C at atmospheric pressure with no carrier gas (Table II).

CVD using aqueous RuO<sub>4</sub> with the substrate at 150 °C gave pure RuO<sub>2</sub>, but the film morphology was very rough and there were many fracture lines. From pentane solutions of RuO<sub>4</sub>, the film morphology was smooth, but adhesion to glass or silicon substrates was poor. The vapor

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<sup>(13)</sup> This method has been reviewed in ref 2a, where additional references are given.

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<sup>(15)</sup> RuO4 is volatile and toxic. It must be handled in a well-ventilated fumehood. Pure RuO<sub>4</sub> has been reported to explode when heated above 100 °C,<sup>10,11</sup> and it may react violently with organic substances.

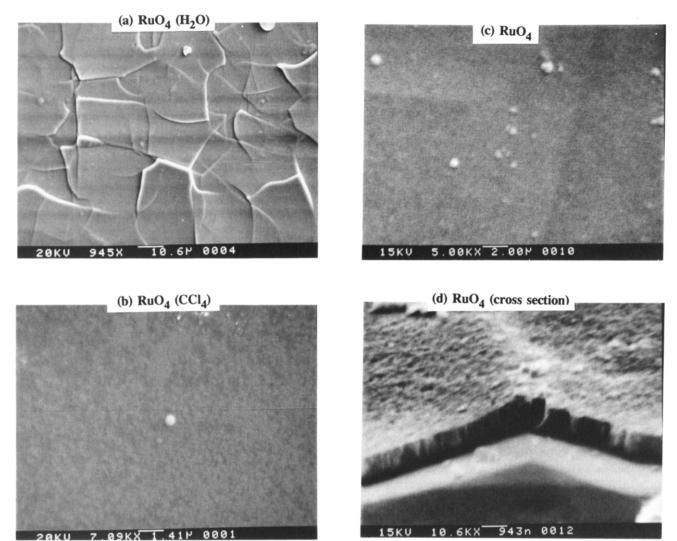


Figure 1. SEM images of films of  $RuO_2$  prepared by atmospheric pressure CVD from  $RuO_4$ : (a) Using aqueous  $RuO_4$  with glass substrate at 150 °C. (b) Using  $RuO_4$  in CCl<sub>4</sub> solution on glass substrate at 220 °C. (c) Using pure  $RuO_4$  on Si(100) substrate at 150 °C. (d) As for (c); cross-section view.

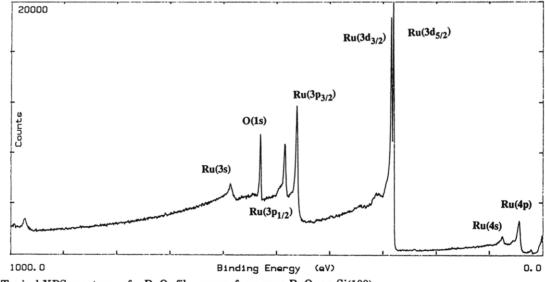


Figure 2. Typical XPS spectrum of a  $RuO_2$  film grown from pure  $RuO_4$  on Si(100).

transport of RuO<sub>4</sub> from CCl<sub>4</sub> solution at room temperature and atmospheric pressure was found to be much slower than from aqueous solution but, using a carrier gas of air or nitrogen, films could be grown with the substrate heated to 220 °C. Low-pressure CVD was less satisfactory since much of the  $RuO_4$  failed to decompose even with the substrate at 250 °C. The film morphology and adhesion were both excellent in the films grown from CCl<sub>4</sub> solution but the films contained significant impurities of chloride (Table I). Using pure RuO<sub>4</sub> as precursor, films were grown

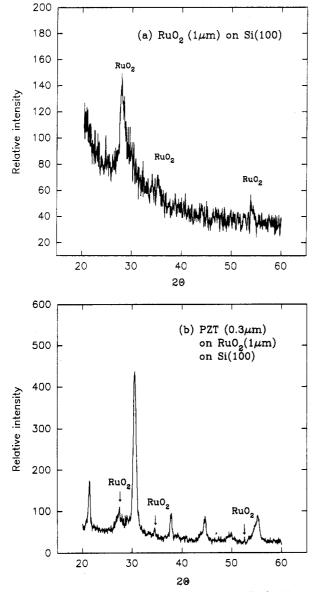


Figure 3. X-ray diffraction patterns: (a) for  $RuO_2$  film on Si(100); (b) for a lead zirconate titanate (Zr/Ti = 40/60) ferroelectric film deposited on a  $RuO_2$  film.  $RuO_2$  peaks are noted.

most easily using a horizontal hot-wall CVD reactor at atmospheric pressure with the substrate at 150 °C. Under these conditions, the film morphology, adhesion and purity were all excellent (Table II). In a typical experiment, a film of 1- $\mu$ m thickness was deposited on a silicon wafer in 2 h with the precursor at room temperature. Since the volatility of the precursor is high, the deposition rate could be greatly increased, and a 1- $\mu$ m film could be grown in a few minutes by using a partial vacuum or by warming the precursor reservoir.

The film morphologies were studied by SEM and typical images are shown in Figure 1. Films prepared from aqueous RuO4 were dull black in color and SEM analysis indicated that the films were cracked and inhomogeneous (Figure 1a). However, films prepared from CCL solution appeared as shiny dark blue mirrors and SEM analysis confirmed their smooth, homogeneous nature (Figure 1b). Similarly, films prepared from pure RuO<sub>4</sub> were of high quality (Figure 1c,d); they adhered well to all of the substrates tested and a film on glass was not etched by acid or base. The XPS spectra of all films showed the Ru  $(3d_{5/2})$  peak at 280.6 eV in good agreement with authentic  $RuO_2$  (Figure 2). The oxygen (1s) chemical shift was 530.1 eV, and for CVD from pure RuO<sub>4</sub>, the Ru:O ratio was typical of bulk  $RuO_2$  (Tables I and II; note that oxygen analyses are always slightly higher than the theoretical values due to surface effects). However, for CVD from aqueous RuO<sub>4</sub>, the RuO<sub>2</sub> film obtained at 150 °C gave a Ru/O ratio much lower than the expected value of 1:2. It is likely that the RuO<sub>2</sub> film is hydrated, perhaps RuO<sub>2</sub>·H<sub>2</sub>O which is a black solid obtained previously by reaction of RuO<sub>4</sub> with H<sub>2</sub>, and which undergoes dehydration only at 587 °C.10 The XPS spectra of films of RuO<sub>2</sub> obtained from aqueous RuO<sub>4</sub> contained an additional, partially resolved band due to O(1s) at 531-532 eV, which may be due to water in the  $RuO_2$ .

The electrical conductivity of the RuO<sub>2</sub> films was measured using gold electrodes evaporated on the face of the sample. The resistivity of a film of thickness approximately 1  $\mu$ m is approximately 10<sup>-2</sup>  $\Omega$  cm.

To assess the application of CVD RuO<sub>2</sub> films for ferroelectric device purposes, lead zirconate titanate films 0.3  $\mu$ m thick were prepared on a 1- $\mu$ m-thick RuO<sub>2</sub> layer using a sol-gel technique.<sup>4</sup> The sample was fired in air on a hot plate at 400 °C for 10 min and then annealed to a fully crystallized state by rapid thermal processing in an AG Associates HeatPulse 410 Rapid Thermal Processor for 30 s at 650 °C. The resulting films were crack-free and had crystallized readily to the perovskite phase (Figure 3). The figure also shows the characteristic diffraction peaks of RuO<sub>2</sub>. After evaporating gold upper electrodes, a ferroelectric hysteresis loop was achieved using standard methods<sup>16</sup> with a remanent polarization of  $30 \,\mu C \, cm^{-2}$  and a coercive field of 150 kV cm<sup>-1</sup>. While these values are not yet optimized, this performance indicates that  $RuO_2$ prepared by CVD has promise as a substrate for ferroelectric thin films. This is likely due both to the ability of RuO<sub>2</sub> to sustain its conductivity during PZT processing and to enhance the nucleation of the perovskite phase in PZT.

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